Anomalous X-ray Diffraction Studies for Photovoltaic Applications

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ABSTRACT

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Anomalous X-ray Diffraction (AXRD) has become a useful technique in characterizing bulk and nanomaterials as it provides specific information about the crystal structure of materials. In this project we present the results of AXRD applied to materials for photovoltaic applications: ZnO loaded with Ga and ZnCo$_2$O$_4$ spinel. The X-ray diffraction data collected for various energies were plotted in Origin software. The peaks were fitted using different functions including Pseudo Voigt, Gaussian, and Lorentzian. This fitting provided the integrated intensity data (peaks area values), which when plotted as a function of X-ray energies determined the material structure.

For the first analyzed sample, Ga was not incorporated into the ZnO crystal structure. For the ZnCo$_2$O$_4$ spinel Co was found in one or both tetrahedral and octahedral sites.
INTRODUCTION

The use of anomalous X-ray diffraction (AXRD) provides element and site specific information for the crystal structure of a material. This technique lets us correlate the structure to the electronic properties of the materials as it allows us to probe precise locations of cations in the spinel structure. What makes it possible is that in AXRD the diffraction pattern is measured at a number of energies near an X-ray absorption edge of an element of interest. The atomic scattering strength of an element varies near its absorption edge and hence the total intensity of the diffraction peak changes by changing the X-ray energy. Thus AXRD provides element specific structural information. This method can be applied to both crystalline and liquid materials. One of the advantages of AXRD in crystallography experiments is its sensitivity to neighboring elements in the periodic tables. This method is also sensitive to specific crystallographic phases and to a specific site in a phase [1].

The main use of AXRD in this study is for transparent conductors (TCs) analysis. TCs are considered to be important materials because of their efficiency and low risk of environmental pollution. These materials are important to solar cells as a result of their remarkable combination of optical and electrical properties, including high electrical conductivity and high optical transparency in the spectrum of visible light [2][3]. TCs provide a transparent window, which allows sunlight to pass through while also allowing electricity to conduct out of the cell.

Spinel materials have the chemical form $AB_2O_4$, and are made of a face-centered cubic (FCC) lattice of oxygen anions and cations in specific interstitial sites (see Figure 1,2) [4] [5]. A normal spinel has all $A$ cations on tetrahedral sites and $B$ cations on octahedral sites. In contrast; an inverse spinel has the $A$ and half of the $B$ cations on octahedral sites and the other half of the $B$ cations on tetrahedral sites; a mixed spinel lies between (see Figure 3). In the spinel structure, 8
of 64 possible tetrahedral sites and 16 of 32 possible octahedral sites are filled. Normal spinels have particularly high conduction as the linear octahedral chains of B cations likely serve as conduction paths [2].

In this paper we present how the data obtained with AXRD is used to analyze TCs properties as they apply to photovoltaic applications. One of the materials used for this analysis is zinc oxide. It has been loaded with 5% and 10% of Ga, which has an absorption edge of 10367 eV. The peak (100) was measured for the zinc oxide loaded with 10% Ga. In the case of 5% Ga, we measured peaks (100) and (101). With the information provided by the AXRD we can identify if Ga is being incorporated in the ZnO crystal structure. The analysis of 311 plane in the ZnCo$_2$O$_4$ spinel shows if Co is in tetrahedral or octahedral site.

**MATERIALS AND METHOD**

*Theoretical Model*

In a diffraction experiment, incident X-rays are directed toward a sample of interest, and are deflected by the atoms in the sample. The waves that are diffracted from different atoms can interfere with each other and consist of sharp interference maxima (peaks) with the same symmetry as in the distribution of atoms [4]. The atomic distance is an important fact in this analysis because it is directly related to the peaks in an X-ray diffraction patterns. For a given set of lattice planes with an inter-planar distance of \( d \), the condition for a diffraction (peak) to occur can be simply written as which is known as the Bragg's law.

\[
n\lambda = 2d \sin \theta \quad (1)
\]
In this equation, $\lambda$ is the wavelength of the X-ray, $\theta$ is the scattering angle, and $n$ is an integer representing the order of the diffraction peak (see Figure 4). It represents the scattering of radiation from two crystal planes of a crystal. These planes behave like half-reflecting mirrors. There is a difference of $2d\sin\theta$ in the pathlength travelled by the two beams of radiation where $d$ is the perpendicular distance between the planes (the path difference is marked in green and the expanded construction alongside makes the geometry more clear). As the angle of reflection is changed so does the difference in pathlength travelled by the two beams. When the path difference is equal to an integer number of wavelengths the two beams will reinforce one another and when it is an integral number of half wavelengths the two waves will interfere destructively with one another. The intensity of the total reflected radiation will vary sinusoidally with $\theta$ [6][7]. Understanding and application of Bragg’s Law can provide much useful information about various samples. It can identify the lattice parameter and phase of the material. Proper understanding of the structure factor can yield more information, particularly when combined with X-ray absorption for AXRD measurements.

$$F_{(h,k,l)} = \sum_{n=1}^{\text{atoms}} f_n(E) e^{(2\pi i)(hx_n + ky_n + lz_n)} \quad (2)$$

$$f_n = f_0(Q) + f'(E) + if''(E) \quad (3)$$

In this equation $f_n$ represents the atomic scattering factor, $x_n$, $y_n$, and $z_n$ are the fractional positions of the $n^{th}$ atom, $f_0(Q)$ is the normal factor (E independent), $f'(E)$ is the anomalous (E dependent), and $f''(E)$ is the absorption (E dependent). The $f_n$ varies near X-ray absorption edge, producing a total intensity change. One fact to be considered about $f_n$ is that it depends on
oxidation state of the element [2]. Calculating the scattering factor let us know the intensity value since both are proportional.

\[ I_{hkl} = \left| F_{hkl} \right|^2 \]  

(4)

In this equation, the intensity, \( I \), is equal to the square of the absolute value of the scattering factor. For this reason, if the X-ray energy near the absorption edge is varied, then total intensity change.

Data Analysis

The software used to develop the analysis was Origin. Once the data was collected at SSRL beamline 2-1, we used Origin to fit peaks and obtain their integrated intensity (peaks area). As peaks are a combination of Gaussian and Lorentzian functions, we used the Pseudo Voigt function for fitting. Compared with Gaussian and Lorentzian functions, Pseudo Voigt gave us the lowest standard errors in the fitting. For each data set we selected some scans to make an average of the \( w_G \), \( w_L \), and \( m_\mu \). The reason we used the average values was to keep the peak shape constant for all X-ray energies.

When the peaks are fitted, we proceed to plot the obtained peak area values as a function of X-ray energies. If we see a feature at an energy corresponding to the absorption edge of a particular element, then that element is present at that particular crystallographic phase. In our case, we are looking for Ga in ZnO, which has an absorption edge at 10367 eV, and for Co in ZnCo$_2$O$_4$ spinel, which has an absorption edge at around 7709 eV. The presence of the cation in
the graph tells us about the material structure. We can determine if the atoms are in tetrahedral or octahedral site in the compound.

RESULTS

The average values obtained using Pseudo Voigt function were 0.5835 for \( w_G \), 1.2969 for \( w_L \), and -0.0196 in \( m_{\mu} \). In the case of the Gaussian and Lorentzian fitting, the average values for each \( w \) were 0.4976 and 0.6196 respectively. For the ZnO loaded with Ga, the data obtained showed a smooth curve in the plot of area as a function of X-ray energies. It did not show any dip at the Ga-absorption edge (see Figures 5,6,7).

Figure 4 shows the fitting for the spinel \( \text{ZnCo}_2\text{O}_4 \) in 311 geometry. There is a big change around 7730 eV, which means that cobalt is present in the sample because its absorption edge is very close to that found in the fitting (7709 eV). The 311 plane probes both tetrahedral and octahedral sites in the spinel crystal structure, so the "dip" seen near the Co absorption edge means that Co is found in one or both of these sites (see Figures 8,9).

DISCUSSION AND CONCLUSIONS

The absorption edge seen for \( \text{ZnCo}_2\text{O}_4 \) determined that Co is present in the sample in tetrahedral and octahedral structure. It means that we can find \( \text{Co}^{2+} \) and \( \text{Co}^{3+} \) in the plane 311. With the obtained results is clear that AXRD is a used technique for the determination of the structure of a material since we were able to obtain the needed data to identify whether the Ga and Co were in tetrahedral or octahedral sites. In the case of the fitting, we can conclude that diffracted peak intensity decreases depending on elements present on diffracting planes. As the energy is varied
through the absorption edge, there is a step function in diffracted intensity. In general, the structural information obtained via diffraction is combined with chemical information obtained via spectroscopy. At an absorption edge, there is a significant change in the atomic scattering factor, which in turn affects the structure factor. Thus by scanning energy while maintaining the Bragg condition for a particular atomic plane, it is possible to determine if certain elements are present based on changes in scattered intensity associated with each element’s absorption edge.

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REFERENCES


Figure 1: (a) Model of spinel $\text{AB}_2\text{O}_4$ structure (b) Model of tetrahedral sites in the spinel (c) Model of octahedral site in the spinel.

Figure 2: In a Face Centered Cubic (FCC) lattice the atoms are arranged at the corners and center of each cube face of the cell.
Figure 3: Representation of normal and inverse spinel of a ZnCo$_2$O$_4$. In the normal spinel Zn atom is in tetrahedral site and both Co atoms are in octahedral sites. For the inverse spinel, Zn is in octahedral site and atoms of Co are in tetrahedral and octahedral sites.

Figure 4: Model of Bragg’s Law. Blue lines indicates the X-ray beamed and diffracted. $\theta$ is the angle formed by the diffraction, $d$ is the distance between the atoms plate.
Figure 5: X-ray diffraction pattern for ZnO. The peak (100) is analyzed using Origin software.

Figure 6: Fitting of peak (100) for scan 90 of ZnO. Red line indicate the fitting using Pseudo Voigt function, green line indicates the fitting using Gauss function, and blue line indicate the fitting using Lorentz function.
Figure 7: Graph of integrated intensity as a function of X-ray energy. Dashed line indicate Ga absorption edge at 10367 eV.

Figure 8: Fitting of peak (311) for scan 90 of ZnCo$_2$O$_4$ spinel. Red line indicate the fitting using Pseudo Voigt function, green line indicates the fitting using Gauss function, and blue line indicate the fitting using Lorentz function.
Figure 9: Graph of integrated intensity as a function of X-ray energy. Dashed line indicate Co absorption edge at 7709 eV.